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# Al-Fe-Ho (Aluminum-Iron-Holmium)

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Recently, Yanson et al. [2001Yan] studied the phase relationships in this ternary system at 500 °C in a narrow region around 10.5 at.% Ho. Here, the structurally-related phases of the types $Th_2Ni_{17}$, $Th_2Zn_{17}$, and $TbCu_7$ occur in close proximity to one another.

## Binary Systems

The Al-Fe phase diagram [1993Kat] shows that the face-centered-cubic (fcc) solid solution based on Fe is restricted by a γ loop. The body-centered-cubic (bcc) solid solution α exists in the disordered *A*2 form, as well as the ordered *B*2 and $D0_3$ forms and extends up to 50 at.% Al at the temperature of interest here (500 °C). In the Fe-Ho system [1982Kub], there are four intermediate phases: $Th_2Ni_{17}$-type hexagonal phase $Ho_2Fe_{17}$ (denoted here 2:17H), $Th_6Mn_{23}$-type cubic phase $Ho_6Fe_{23}$, $PuNi_3$-type rhombohedral phase $HoFe_3$, and $MgCu_2$-type cubic phase $HoFe_2$.

## Ternary Isothermal Section

With starting metals of purity of not less than 99.9%, [2001Yan] melted about 65 alloy compositions in an arc furnace under Ar atm. Three series of compositions were prepared at 9.5, 10.5, and 11.5 at.% Ho and at 2.5 at.% Al intervals up to 50 at.% Al. Some samples were also prepared at 5 and 15 at.% Ho. The samples were given a final anneal at 500 °C for 720 h and quenched in water. The phase equilibria were studied by x-ray powder diffraction.

At 9.5 at.% Ho, the 2:17H phase, the $TbCu_7$ type hexagonal phase 1:7, and the $ThZn_{17}$ type rhombohedral phase 2:17R are stable in the range of 0-27, 30 and 32-37 at.% Al, respectively. At 10.5 at.% Ho, they cover the concentration ranges of 0-20, 22-25, and 30-32 at.% Al, respectively. At 11.5 at.% Ho, only 2:17R is stable in the range 22-30 at.% Al. The influence of the geometrical factors on the structural stability of these phases is discussed by [2001Yan].

Due to the narrow range of the Ho content and the extensive line overlap in the x-ray patterns of the closely-related structures, the identification of the two-phase fields or of the presence of other closely-related structure variants such as the rhombohedral $PrFe_7$ type could not be done. A single plot of the subcell parameters of all the three structures indicated a general increase in the *a* and *c* parameters with increasing Al content, except that the *c* parameter remains practically constant for the 2:17R phase [2001Yan].

The partial isothermal section at 500 °C constructed by [2001Yan] is redrawn in Fig. 1. The ordered forms of Fe-Al bcc phase α are not shown separately from α. The three phases 2:17H, 1:7, and 2:17R are "in equilibrium with the neighboring solid solutions α, $Ho(Al,Fe)_2$ and $Ho(Fe,Al)_{12}$ and the ternary compounds having $ThMn_{12}$ and $MgZn_2$ type structures," [2001Yan]. As seen in Fig. 1, these equilibria are sketched schematically by [2001Yan] without labeling the phase fields. The details are not known.

### References

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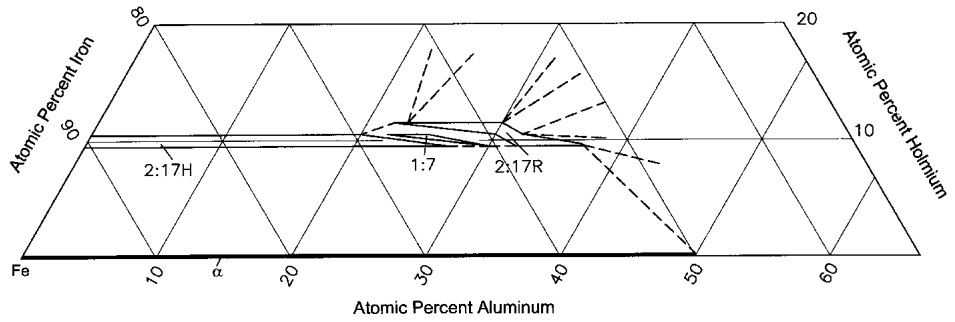


**Fig. 1** Al-Fe-Ho partial isothermal section at 500 °C [2001Yan]